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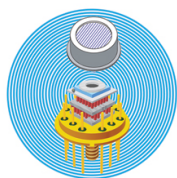
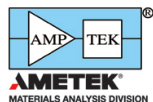
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Origin of plasma-induced surface roughening and ripple formation during plasma etching: The crucial role of ion reflection

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Plasma-induced surface roughening and ripple formation has been studied based on Monte Carlo simulations of plasma-surface interactions and feature profile evolution during Si etching in Cl-based plasmas, with emphasis being placed on the role and effects of ion reflection from microstructural feature surfaces on incidence. The simulation model included the effects of Cl⁺ ion reflection (and/or its penetration into substrates) through calculating the momentum and energy conservation in successive two-body elastic collisions with substrate Si atoms every ion incidence. The “reflection coefficient r_i ” was then further introduced in the model ($0 \leq r_i \leq 1$), representing the fraction of ions incident on surfaces with the reflection/penetration calculation scheme turned on. The coefficient r_i is, in a sense, a measure of the reflection probability for impacts of an ion species onto Si surfaces relative to that for Cl⁺ impacts. Simulations for ion incidence angles of $\theta_i = 0^\circ$, 45° , and 75° onto substrate surfaces with incident energies in the range $E_i = 20\text{--}500$ eV showed that as r_i is slightly decreased from unity, the roughness decreases substantially, and the ripple formation fades away: the roughness remains at the low level of stochastic roughening during etching for decreased $r_i \leq r_i^* \approx 0.95\text{--}0.75$ (the critical r_i^* tends to be lower at higher E_i and θ_i) with no ripple structures at off-normal θ_i . This elucidates that the ion reflection is indispensable in surface roughening and rippling during plasma etching, and their degree relies significantly on the reflectivity of ions. Simulations further showed that at intermediate off-normal $\theta_i = 45^\circ$, the ripple wavelength increases significantly with decreasing r_i , while the increase in amplitude is relatively less significant; thus, sawtooth-like ripple profiles pronounced for $r_i = 1$ tend to be collapsed with decreasing r_i . These effects of reduced ion reflection on plasma-induced surface roughening and ripple formation are discussed in terms of effectively enhanced smoothing due to neutral reactants, which competes with the roughening and rippling caused by ion bombardment. *Published by AIP Publishing.*

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I. INTRODUCTION

Plasma-induced surface roughness at atomistic scale has become an important issue in the fabrication of nanoscale microelectronic and microelectromechanical devices,^{1,2} because it largely affects the variability in device performance.^{3,4} Surface roughening during plasma etching has been studied extensively,^{5–7} including sidewall roughening in pattern transfer^{8–15} and also maskless formation of organized nanostructures such as nanotextures and nanopillars.^{16–21} Correspondingly, several mechanisms have been invoked in

continuum models and Monte Carlo (MC) simulations to reproduce the experiments:^{5,7,18,22–24} stochastic roughening,^{5,7,22–25} geometrical shadowing,²² reemission of neutrals,^{23,24,26} micromasking by inhibitors,^{18,27} and ion reflection.^{5,7,25,28–30} Longitudinal striations or ripplelike structures (called the line edge/width roughness) often formed on feature sidewalls^{8–15} are usually appreciated to arise extrinsically from pattern transfer of the mask edge roughness under the effects of ion shadowing;¹² in practice, they would also arise intrinsically or spontaneously from plasma-surface interactions themselves on feature sidewalls at high off-normal ion incidence, although little mechanistic work has been concerned with surface roughening and rippling in response to ion incidence angle during plasma etching.^{5,7,25,31–33}

The morphological evolution of solid material surfaces at atomistic scale has also been of interest during ion beam sputtering (IBS); in particular, spontaneous or self-organized formation of ordered nanostructures such as periodic dots and ripples in response to ion incidence angle onto surfaces has been extensively studied since decades,^{34–39} as a promising approach to fabricate well-ordered nanostructures.

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The formation and evolution of different surface morphologies in IBS is often assumed to be due to a competition between surface roughening or instability (caused by curvature-dependent sputtering)^{40–43} and smoothing (caused by surface diffusion and/or viscous flow of surface atoms).^{44–47} Several mechanisms have been further invoked to reproduce the experiments, including shadowing,⁴⁸ redeposition,⁴⁹ mass redistribution,^{50–52} and ion reflection/channeling.^{53–56}

Plasma etching and IBS are mechanistically similar in that surface erosion or material removal is affected critically by ion bombardment: the former results primarily from ion-enhanced etching through synergistic interactions of simultaneously incident ions (typically 50–500 eV) and neutral reactants with surfaces, while the latter from physical sputtering only through ion bombardment (typically 10^3 – 10^5 eV). However, mechanistic discussions of the surface morphology evolution are frequently disconnected between the two fields, although similar mechanisms have been invoked in both fields to reproduce the experiments as above; in particular, the role and effects of ion scattering or reflection from feature surfaces on incidence have not been fully understood in both fields, although they are recognized to be key to the formation and evolution of specific geometrical features therein.

In plasma etching, the ion reflection was taken into account in continuum models and MC simulations for feature profile evolution, usually assuming specular reflection from feature sidewalls, to reproduce microtrenches formed at the corner of the feature bottom;^{57–65} a few plasma beam and MC studies of Sawin *et al.* suggested the importance of the effects of ion reflection on surface roughening and rippling;^{5,25,31–33} the ion reflection was also suggested as a possible explanation for the evolution of surface roughness during pulsed plasma etching.²⁸ In IBS, Hauffe suggested that the coarsening of terrace-like (or sawtooth-like) ripple structures is caused by reflected ions that re-impinge on the adjacent facets,⁵³ which was supported in several experiments^{66–69} while not in some other experimental⁷⁰ and continuum model⁷¹ studies; Bradley and Harper gave little more than comments on the effects of ion reflection on the ripple formation, in their pioneering theoretical study based on continuum equations;⁴⁰ a few classical molecular dynamics (MD) simulations were concerned with the step-edge sputter yields at high off-normal ion incidence arising from the ion reflection and re-impingement and with their effects on the ion surface channeling and ripple formation thereat.^{54–56}

We have investigated surface roughening and ripple formation during Si etching in Cl-based plasmas;^{7,72–78} our MC-based three-dimensional atomic-scale cellular model (ASCeM-3D), mechanistically including the effects of ion reflection, simulated the roughening and rippling in response to ion incidence angle θ_i .^{72,73} Moreover, the experiments of inductively coupled plasma (ICP) etching of Si in Cl_2 with different ion incident energies of $E_i \approx 20$ –500 eV ($\theta_i = 0^\circ$) exhibited two modes of surface roughening: ^{74,75} roughening mode at low E_i (<200–300 eV) and smoothing or non-roughening mode at higher E_i . A comparison of the experiments and MC simulations with the help of MD^{76,77} and plasma diagnostics⁷⁴ suggested that the roughening-smoothing transition observed

is characterized by reduced effects of the ion reflection, caused by a change in the predominant ion flux due to that in plasma conditions.⁷⁴ Further experiments using sheath control plates to achieve the off-normal ion incidence onto substrate surfaces ($E_i \approx 500$ eV; $\theta_i \approx 40^\circ, 80^\circ$) first demonstrated the formation of well-defined periodic surface ripples by plasma etching (though in the smoothing mode for normal $\theta_i = 0^\circ$).⁷⁸

In this paper, we elucidate quantitatively the crucial role of ion reflection in surface roughening and rippling during plasma etching, based on MC-based ASCeM-3D simulations of Si etching in Cl-based plasmas with $E_i \approx 20$ –500 eV at $\theta_i = 0^\circ$ – 85° . The model includes the effects of ion reflection from feature surfaces on incidence (and/or its penetration into substrates) through calculating the momentum and energy conservation in successive two-body elastic collisions with substrate atoms every ion incidence (not assuming specular reflection). In this study, the “reflection coefficient r_i ($0 \leq r_i \leq 1$)” is further introduced in the model, representing the fraction of ions incident on surfaces with the reflection/penetration calculation scheme turned on; preliminary results for normal $\theta_i = 0^\circ$ were in part described previously.^{7,74,75} A quantitative understanding, under what conditions the reduced ion reflection ($r_i < 1$) results in surface smoothing/non-roughening and under what conditions it retains the roughening and rippling, would be of great technological as well as fundamental importance. Moreover, we discuss the ripple topography and dynamics (wavelength, amplitude, profile, and propagation velocity) at intermediate off-normal $\theta_i = 45^\circ$ depending on the degree of ion reflection. These effects of reduced ion reflection on plasma-induced surface roughening and ripple formation are considered as effectively equivalent to those of enhanced smoothing due to neutral reactants, which competes with the roughening and rippling caused by ion bombardment.

II. MODEL

The ASCeM-3D is concerned with plasma-surface interactions and feature profile evolution during Si etching in Cl-based plasmas, which has been detailed previously.^{7,72,73} Briefly, the model takes into account the transport and surface reaction kinetics of Cl^+ ions, Cl and O neutrals, and etch/sputter products and byproducts of SiCl_x and SiCl_xO_y in microstructural features. The simulation domain consists of small cubic cells of atomic size ($L = \rho_s^{-1/3} = 2.7 \text{ \AA}$ with $\rho_s = 5.0 \times 10^{22} \text{ cm}^{-3}$, the atomic density of Si substrates), where the evolving interfaces are represented by removing Si atoms from and/or allocating them to the substrate surface cells concerned. The transport of ions and neutrals is analyzed based on single-particle trajectories with periodic boundary conditions in horizontal directions, where the transport is calculated every movement of step L , taking into account the geometrical shadowing effects of the feature and those of surface reemission for neutrals. The particles are assumed to move straight without collisions with other particles in vacuum; then, the particles are assumed to reach the surface if there is a Si atom in any of the 26 cells neighboring the cell in which the particle concerned is. The local surface normal and thus the local angle θ of incidence (relative to it)

on feature surfaces is calculated constantly using the extended four-point technique for $5 \times 5 \times 5$ neighboring cells (125 cells in total) at around the substrate surface cell that the particle reaches. Note that for primary incident ions, $\theta = \theta_i$ on flat or completely planar substrate surfaces, while generally $\theta \neq \theta_i$ on local surfaces of microstructural features.

Surface chemistry and kinetics include surface chlorination, chemical etching, ion-enhanced etching, sputtering, surface oxidation, redeposition of etch/sputter products desorbed from feature surfaces being etched, and deposition of etch/sputter byproducts coming from the plasma. The ion-related etch/sputter yields depending on ion incident energy E_i and angle θ are approximated by the functional form of the universal energy dependence:⁷⁹ $Y(E_i, \theta) = A (E_i^{1/2} - E_{th}^{1/2}) g(\theta)$ at $E_i \geq E_{th}$, where Y is the total number of Si atoms desorbed from surfaces per ion impact, A is a proportional constant or scaling factor (taken to rely on surface conditions^{7,72,73}), and E_{th} is the threshold energy concerned. In plasma etching where the ion-enhanced etching reactions dominate over sputtering, the etch yield is assumed to be maximum at normal $\theta = 0^\circ$, while the sputter yield peaks at off-normal $\theta \approx 60^\circ - 70^\circ$.^{7,72}

The model also takes into account the ion reflection from feature surfaces on incidence into vacuum and/or the ion penetration into substrates.^{80,81} The ion reflection/penetration is analyzed every ion incidence, using 3D MC calculation of single-particle Cl^+ ion trajectories according to the momentum and energy conservation through successive two-body elastic collisions with the most neighboring substrate Si atoms;^{7,72,73} the ions reflected from the surface into vacuum (in some cases at high off-normal incidence, scattered out of the surface after channeling) are taken to move further toward another surface of the feature or go out of the simulation domain, similarly to neutrals reemitted. The interaction potential for Cl-Si employed is the simple, repulsive Ziegler-Biersack-Littmark (ZBL) type.^{82,83}

Simulations were made for square substrates $W = 50$ nm on a side having initially flat or completely planar surfaces

(their root-mean-square roughness $\text{RMS} = 0$ at $t = 0$), with different ion incident energies of $E_i = 20 - 500$ eV and incidence angles of $\theta_i = 0^\circ, 45^\circ$, and 75° onto substrate surfaces (relative to the substrate surface normal). The following conditions were then assumed as in high-density plasma etching environments (unless otherwise stated):^{74,75} a directionally incoming ion flux $\Gamma_i^0 = 1.0 \times 10^{16} \text{ cm}^{-2} \text{ s}^{-1}$, ion temperature $k_B T_i = 0.5$ eV ($R = E_i/k_B T_i$ reflects the directionality or the angular distribution of incoming ion fluxes), isotropically incoming neutral reactant-to-ion flux ratio $\Gamma_n^0/\Gamma_i^0 = 100$, and neutral temperature $T_g = 500$ K in the absence of incoming etch inhibitors such as oxygen and byproducts ($\Gamma_o^0 = \Gamma_p^0 = 0$); in addition, the sticking probability $S_q = 0.05$ was assumed for redeposition of etch/sputter products, along with the dopant concentration $N_e = 1.0 \times 10^{18} \text{ cm}^{-3}$ and surface temperature $T_s = 320$ K of substrates. Note that the effects of N_e and T_s are included in the model via the known reaction probability for Cl-atom chemical etching of Si depending on N_e and T_s (obtained from experiments of Si etching in heated Cl_2 gases).⁸⁴ The etching or plasma exposure time was 1–2 min in this study [ion fluence $\Phi = \Gamma_i^0 \times \text{time} = (0.6 - 1.2) \times 10^{18} \text{ cm}^{-2}$], which is typical in industry.

III. RESULTS AND DISCUSSION

A. Role of ion reflection in roughening and rippling

Figure 1 shows typical surface features of Si at $t = 60$ s after the start of etching in Cl_2 plasma (angle view, $50 \times 50 \text{ nm}^2$), simulated with different ion reflection coefficients in the range $r_i = 1 - 0$ for $E_i = 100$ eV at $\theta_i = 0^\circ, 45^\circ$, and 75° . The corresponding etch rate and rms surface roughness are shown in Fig. 2 as a function of r_i , together with the ion-enhanced etch rate $\text{ER}_{\text{ion}} = Y_{\text{SiCl}_4}(E_i, \theta) \Gamma_i^0 / \rho_s$ of Si surfaces chlorine saturated bombarded by Cl^+ ions (assuming $\theta = \theta_i$).^{85,86} The “reflection coefficient r_i ($0 \leq r_i \leq 1$)” is defined as the fraction of ions incident on surfaces with the reflection/penetration calculation scheme turned on, or the coefficient $(1 - r_i)$ is as the fraction of incident ions with the

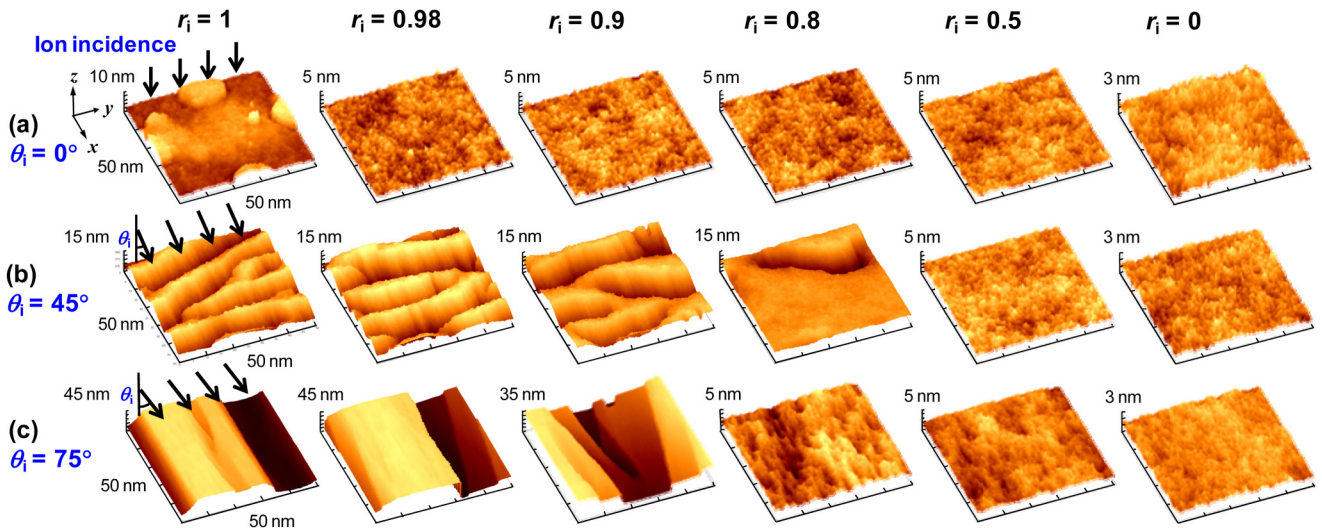


FIG. 1. Typical surface features of Si at $t = 60$ s after the start of etching in Cl_2 plasma (angle view, $50 \times 50 \text{ nm}^2$), simulated with different ion reflection coefficients in the range $r_i = 1 - 0$ for an ion incident energy of $E_i = 100$ eV at different incidence angles of $\theta_i =$ (a) 0° , (b) 45° , and (c) 75° onto substrate surfaces. The coordinate system (x, y, z) used in the model is inset in (a).

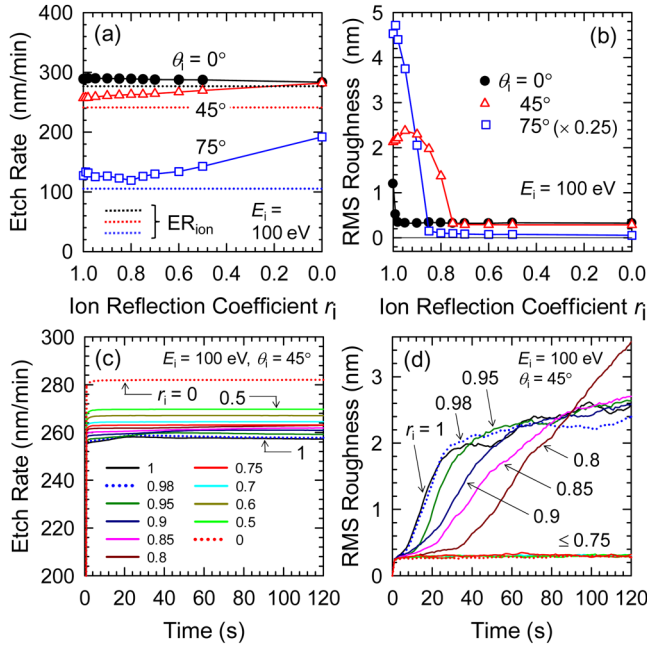


FIG. 2. (a) Etch rate and (b) rms surface roughness of Si at around $t=60$ s as a function of ion reflection coefficient in the range $r_i=1-0$ for $E_i=100$ eV at different $\theta_i=0^\circ$, 45° , and 75° , obtained from simulations as in Fig. 1. Also shown in (a) for reference is the ion-enhanced etch rate $ER_{ion} = Y_{SiCl_4}(E_i, \theta) \Gamma_i^0 / \rho_s$ of Si surfaces chlorine saturated (assuming $\theta = \theta_i$). (c) Etch rate and (d) rms roughness of Si as a function of time in the range $0 \leq t \leq 120$ s for different $r_i=1-0$ with $E_i=100$ eV at $\theta_i=45^\circ$, obtained from simulations as in Fig. 1(b). The data shown are the averages extracted from several runs of the simulation.

reflection/penetration calculation turned off; in other words, at least $(1 - r_i) \times 100\%$ of the incident ions are taken to stop and deposit their kinetic energy at the point of surfaces on incidence without reflection/penetration. Note that in ASCeM-3D, the ion reflection/penetration is calculated every ion incidence (as mentioned in Sec. II), giving internally the reflection probability $R_i(E_i, \theta)$ depending on ion incident energy and incidence angle along with the distributions of reflected energies and angles;^{7,63} e.g., the reflectivity R_i for 100-eV Cl^+ impacts onto bare Si surfaces is evaluated to increase from 0.01 at $\theta_i=0^\circ$ to ≥ 0.46 at $\theta \geq 70^\circ$. Then, the

probability of ion reflection in this study is assumed to be effectively given by $\mathcal{R}_i(E_i, \theta) = r_i \times R_i(E_i, \theta)$, where $0 \leq r_i$, R_i , $\mathcal{R}_i \leq 1$, and the coefficient r_i may be assumed to be a measure of the reflectivity for ion species on Si surfaces (a second condition of interest) relative to that for Cl^+ on Si (base condition), as further discussed later with respect to Fig. 3. Note that the ions assumed under a second condition are virtual ones, in a sense, having the same mass, interaction potential, and thus collision dynamics (such as incident energy and angle dependences of the reflection probability, and energy and angle distributions of reflected atoms/molecules) as Cl^+ except for the magnitude of the reflectivity.

Simulations with conventional $r_i=1$ ^{7,72,73} exhibit the random roughness at normal incidence ($\theta_i=0^\circ$); in contrast, they exhibit sawtooth-like ripples with their wave vector oriented parallel (crests/troughs elongated perpendicular) to the direction of ion incidence at intermediate off-normal angles ($15^\circ < \theta_i < 60^\circ$), while striations or ripplelike structures with the wave vector perpendicular (crests/troughs parallel) to it at high off-normal angles ($70^\circ < \theta_i < 85^\circ$).

As the reflection coefficient r_i is slightly decreased from unity, the roughness is reduced and the ripple structures tend to fade away: the rms roughness decreases substantially down to the low level of stochastic roughening (RMS < 0.3 nm, slightly larger at higher E_i and lower θ_i) for decreased $r_i \leq r_i^* \approx 0.95, 0.75$, and 0.8 at $\theta_i=0^\circ, 45^\circ$, and 75° , respectively, showing no pronounced ripple structures at off-normal θ_i . Note that the low-level stochastic roughening is due to the uniformity of incident fluxes of ions and/or neutral etchants on surfaces at microscale, and thus its degree depends slightly on plasma conditions such as $E_i, \theta_i, \Gamma_n^0/\Gamma_i^0$, and T_s .⁷ The rms roughness does not scale with the etch rate, which tends to be increased slightly with decreasing r_i owing to the increased ion energy deposited on surfaces on incidence; the etch rate is larger at $\theta_i=0^\circ$ and 45° than at $\theta_i=75^\circ$, reflecting the angle dependence of the etch yield $Y_{SiCl_4}(E_i, \theta)$ taken in the model. In simulations, as also shown in Fig. 2 partly for $\theta_i=45^\circ$, the rms roughness for $r_i > r_i^*$ exhibits a gradual increase with time to reach quasi-steady state in $t < 120$ s at $\theta_i=0^\circ$ and 45° , while it exhibits a continuous increase during etching at $\theta_i=75^\circ$;^{7,72,73}

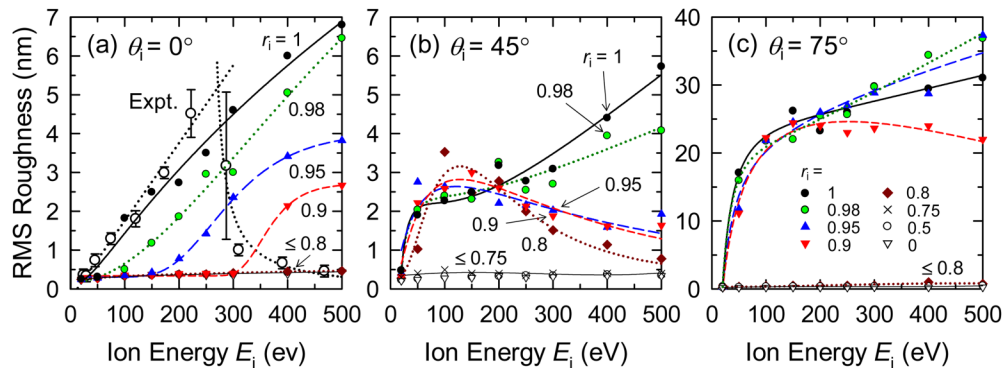


FIG. 3. RMS surface roughness of Si at around $t=120$ s as a function of ion incident energy in the range $E_i=20-500$ eV, obtained from simulations of Si etching in Cl_2 plasma for different reflection coefficients of $r_i=1-0$ at different incidence angles of $\theta_i=(a) 0^\circ$, (b) 45° , and (c) 75° . Note that the low-energy conditions of $E_i=20$ eV at any θ_i and $E_i=50$ eV at $\theta_i=0^\circ$ do little roughening on their own even for $r_i=1$. The data shown are the averages extracted from several runs of the simulation, and the solid, dashed, and dotted lines in the respective figures are for guiding the eyes only. Additionally shown in (a) for comparison is the rms roughness experimentally obtained in ICP Cl_2 plasma etching of Si.⁷⁴

in contrast, for $r_i \leq r_i^*$ at any θ_i , the roughness retains the low level during etching after only a little increase at the initial stage; in addition, the etch rate reaches quasi-steady state in $t < a$ few seconds soon after the start of etching, irrespective of r_i and θ_i .

These results clearly indicate the crucial role of ion scattering or reflection from feature surfaces on incidence in surface roughening and ripple formation during plasma etching: it is indispensable therein, and the degree of roughening and rippling relies significantly on the reflectivity of ions. In effect, irrespective of macroscopic incidence angle θ_i onto substrate surfaces, the ions are expected to be often incident on sidewalls of roughness features at locally high off-normal angles θ ; such off-normally incident ions are then reflected therefrom and tend to re-impinge on the neighboring sidewalls and/or bottom surfaces of the feature, give rise to significant momentum and energy transfer thereto, cause enhanced etching of roughness features, and thus enhance their degree of roughness to its lateral as well as vertical extent and also induce the ripple formation.

Figure 3 shows the rms surface roughness of Si at around $t = 120$ s as a function of ion energy in the range $E_i = 20$ –500 eV, obtained from simulations of Si etching in Cl_2 plasma for different reflection coefficients of $r_i = 1$ –0 at $\theta_i = 0^\circ$, 45° , and 75° . The rms roughness for conventional $r_i = 1$ is increased with increasing E_i irrespective of θ_i , where low-energy conditions of $E_i = 20$ eV at any θ_i and $E_i = 50$ eV at $\theta_i = 0^\circ$ do little roughening on their own even for $r_i = 1$; note that this is not inconsistent with the experiments of atomic layer etching of Si,^{1,2} where low-energy Ar^+ ions impact on chlorinated surfaces ($E_i \approx 50$ eV, $\theta_i = 0^\circ$). The increase in roughness with increasing E_i has been observed in some experiments of plasma etching of Si in Cl_2 ,⁸⁷ while some other experiments showed a decrease in roughness with increasing E_i during plasma etching of Si in Cl_2 ⁸⁸ and SF_6 .^{89,90}

The decrease in roughness (or the smoothing/non-roughening) with reduced ion reflection tends to be less significant at increased E_i and θ_i : the higher the E_i and the θ_i , the smaller is the critical r_i^* giving the rms roughness decreased down to the low level of stochastic roughening (RMS < 0.3–0.5 nm); concretely, $r_i^* \approx 0.95$ –0.8 (decreasing significantly with increasing E_i) at normal $\theta_i = 0^\circ$, while $r_i^* \approx 0.8$ –0.75 (almost independent of E_i investigated) at off-normal $\theta_i = 45^\circ$ and 75° . In practice, the effects of ion reflection and thus the effects of reduced ion reflection are expected to be less significant for 2D ripples at $\theta_i = 45^\circ$ and 75° than for 3D random roughness features at $\theta_i = 0^\circ$, because the flux of ions reflected from feature sidewalls and then re-impinging on the neighboring deeper surfaces of the feature^{59,62} is assumed to be smaller for 2D features (such as trenches, opened in one direction) than for 3D features (such as holes/vias, surrounded by walls on all sides). The reduced effects of ion reflection followed by re-impingement are also expected to be effectively less significant at increased E_i , where the increased directionality of incoming ion fluxes is assumed to increase the flux of ions directly incident on the bottom of roughness features, and thus to decrease the flux of ions reflected from feature sidewalls. In addition, the reduced effects of ion reflection at $\theta_i = 75^\circ$ would be slightly

significant as compared to those at $\theta_i = 45^\circ$ ($r^* \approx 0.75$ and 0.8 at $\theta_i = 45^\circ$ and 75° , respectively), presumably owing to the contribution of the ion scattering-caused channeling effects at high off-normal incidence.

The critical r_i^* values at normal $\theta_i = 0^\circ$ indicate that for decreased $r_i \leq 0.8$, the rms roughness retains the low level of stochastic roughening even at high $E_i > 400$ eV for $\theta_i = 0^\circ$. This is consistent with our ICP plasma etching experiments of Si in Cl_2 exhibiting the roughening-smoothing transition at increased $E_i > 200$ –300 eV,^{7,74,75} as additionally shown in Fig. 3(a): the rms roughness in experiments increases with increasing E_i at low $E_i < 200$ –300 eV (roughening mode, scaling with the etch rate), which is well reproduced by the simulations with conventional $r_i = 1$; on the other hand, at further increased E_i , it decreases substantially with E_i down to the low level (smoothing mode, not scaling with the etch rate), which is not reproduced by them, but reproduced by simulations with decreased $r_i \leq 0.8$. Note that in our experiments, the etch rate was observed to increase with increasing E_i over the E_i range investigated (both roughening and smoothing modes), which is reproduced easily by the simulations with any $r_i = 1$ –0 [the numerical etch rate $\text{ER} \approx \text{ER}_{\text{ion}} \propto E_i^{1/2}$ is relatively insensitive to r_i as partly seen in Figs. 2(a) and 2(c)].^{7,74}

Correspondingly, the predominant ion flux was found to change from reactive $\text{Cl}_2^+/\text{SiCl}_3^+$ at low E_i (base condition) to depositive SiCl^+ at increased E_i (a second condition). In effect, as the ion energy E_i was increased in experiments,^{7,74} the fluxes of ionized etch products/byproducts SiCl_x^+ ($x = 0$ –4) were observed to increase owing to the increase in etch rate, while those of feed gas ions Cl_x^+ ($x = 1, 2$) were to decrease partly owing to the reduced partial pressures or concentrations of feed gases under operating conditions of constant pressure; as a result, the predominant ion flux during etching was Cl_2^+ (followed by SiCl_3^+ , SiCl^+ , and SiCl_4^+) at low $E_i < 30$ eV, SiCl_3^+ (followed by Cl_2^+ , SiCl^+ , and SiCl_4^+) at $30 < E_i < 50$ eV, SiCl_3^+ (followed by SiCl^+ , Cl_2^+ , and SiCl_4^+) at $50 < E_i < 120$ eV, and then SiCl^+ (followed by SiCl_3^+ , SiCl_4^+ , and Cl_2^+) at further increased $E_i > 120$ eV. At high $E_i > 300$ eV, the SiCl^+ , SiCl_3^+ , SiCl_4^+ , and Cl_2^+ fluxes were relatively 1, < 0.3, < 0.06, and < 0.02, respectively. Such an aspect of the predominant ionic species (SiCl_x^+ and not Cl_x^+) during plasma etching of Si in Cl-based plasmas has been predicted by recent modeling simulations,⁹¹ being attributable partly to the total electron-impact ionization cross section for SiCl_x ($x = 1$ –4) molecules, which is about 2–3 times larger than that for Cl_2 .

According to MD simulations of Cl_x^+ ($x = 1, 2$) and SiCl_x^+ ($x = 0$ –4) ion impacts on Si surfaces with $E_i = 20$ –500 eV at $\theta = 0^\circ$ – 85° ,^{7,77} incident molecular ions often fragment or dissociate upon impact. In Cl^+ , Cl_2^+ , SiCl^+ , and SiCl_3^+ incidences, e.g., the atoms/molecules scattered or reflected directly from the surface on incidence during impact are Cl for Cl^+ impact, Cl and Cl_2 for Cl_2^+ impact, Cl, Si, and SiCl for SiCl^+ impact, and Cl, Cl_2 , Si, SiCl, SiCl_2 , and SiCl_3 for SiCl_3^+ impact, including fragments upon impact. The fraction of incident molecular ions that are dissociated upon impact increases with increasing ion incident energy E_i , and for a given E_i , it peaks at around $\theta \approx 70^\circ$. In these situations,

the reflection probability $R_i(E_i, \theta)$ for atomic Cl^+ impact is generally calculated as the total number of reflected Cl atoms divided by that of incident ones, while the probability for molecular Cl_2^+ , SiCl^+ , and SiCl_3^+ impacts is calculated in our MD simulations taking into account the effects of dissociation upon impact, as will be detailed elsewhere.⁷⁷

The reflection probability $R_i(E_i, \theta)$ thus calculated tends to be lower for depositive SiCl^+ than for reactive Cl^+ , Cl_2^+ , and SiCl_3^+ ,^{7,77} e.g., the reflectivity R_i for $E_i = 300$ eV ion impacts increases from 0.061 at $\theta = 0^\circ$ to ≥ 0.54 at $\theta \geq 70^\circ$ for Cl^+ , from 0.11 at $\theta = 0^\circ$ to ≥ 0.73 at $\theta \geq 70^\circ$ for Cl_2^+ , from 0.14 at $\theta = 0^\circ$ to ≥ 0.83 at $\theta \geq 70^\circ$ for SiCl_3^+ , and from 0.063 at $\theta = 0^\circ$ to ≥ 0.67 at $\theta \geq 70^\circ$ for SiCl^+ ; note that R_i is lower for higher E_i owing to increased penetration at normal and off-normal incidence angles of approximately $\theta < 70^\circ$, while it tends to be higher for higher E_i at high off-normal $\theta > 70^\circ$. Thus, the ratio of the reflectivity R_i for SiCl^+ to that for $\text{Cl}_2^+/\text{SiCl}_3^+$ is on the order of $\chi_i \approx 0.8\text{--}0.9$ at $\theta > 70^\circ$, e.g., for $E_i = 300$ eV (χ_i tends to be larger for higher E_i), which reflects the smoothing mode in experiments ($\theta_i = 0^\circ$) reproduced by ASCeM-3D with decreased r_i as in Fig. 3(a). In addition, the smaller $r_i^* \approx 0.8\text{--}0.75$ at off-normal $\theta_i = 45^\circ$ and 75° (as compared to $r_i^* \approx 0.95\text{--}0.8$ at normal $\theta_i = 0^\circ$) is also consistent with our recent experiments of Si etching in ICP Cl_2 plasmas using sheath control plates:⁷⁸ the ripple formation was demonstrated under conditions of $E_i \approx 500$ eV with $\theta_i \approx 40^\circ$ and 80° , where the predominant ions were SiCl^+ with relatively low R_i , giving the smoothing mode for $\theta_i = 0^\circ$.

B. Similarity between reduced ion reflection and enhanced smoothing

Similar smoothing/non-roughening behavior was observed in ASCeM-3D simulations with $r_i = 1$, through increasing the neutral reactant flux ($\Gamma_n^0/\Gamma_i^0 = 50\text{--}1000$)⁷³ and the surface temperature ($T_s = 320\text{--}500$ K),⁹² although it is mechanistically different from the smoothing with reduced ion reflection discussed above in Sec. III A: the roughness is reduced and the ripples tend to fade away as Γ_n^0/Γ_i^0 and T_s are increased, which is also less significant at increased E_i and θ_i . As Γ_n^0/Γ_i^0 is increased for a given ion energy $E_i = 100$ eV (with $T_s = 320$ K),⁷³ e.g., the rms roughness increases (scaling with the etch rate) at low $\Gamma_n^0/\Gamma_i^0 < 50$ irrespective of the incidence angle θ_i . Then, the roughness at $\theta_i = 0^\circ$ peaks at $\Gamma_n^0/\Gamma_i^0 \approx 50$ and then decreases substantially down to the low level of stochastic roughening at $\Gamma_n^0/\Gamma_i^0 > 200$; the roughness at $\theta_i = 45^\circ$ remains almost constant at $\Gamma_n^0/\Gamma_i^0 > 50$ and tends to decrease at $\Gamma_n^0/\Gamma_i^0 > 500$; the roughness at $\theta_i = 75^\circ$ continues to increase and remains almost constant at $\Gamma_n^0/\Gamma_i^0 > 500$. The ripples at $\theta_i = 45^\circ$ are significant in the range $\Gamma_n^0/\Gamma_i^0 \approx 50\text{--}500$, and they tend to fade away at further increased Γ_n^0/Γ_i^0 , where the ripple wavelength is shortest at $\Gamma_n^0/\Gamma_i^0 \approx 100$; the ripples at $\theta_i = 75^\circ$ enlarge significantly at increased $\Gamma_n^0/\Gamma_i^0 > 100$, exhibiting a wide, deep groove. Moreover, as T_s is increased for a given $E_i = 100$ eV (with $\Gamma_n^0/\Gamma_i^0 = 100$),⁹² e.g., the rms roughness decreases substantially down to the low level at $T_s > 360$ K for $\theta_i = 0^\circ$ and at $T_s > 500$ K for $\theta_i = 45^\circ$ and 75° . Correspondingly, the ripples at $\theta_i = 45^\circ$ and 75° tend to fade

away at increased T_s , where the wavelength of the former is increased with increasing T_s . In practice, the surface smoothing with increased Γ_n^0/Γ_i^0 ($\theta_i = 0^\circ$) has been observed in some experiments of plasma etching of Si in SF_6 ,⁸⁹ which showed a decrease in roughness with increasing pressure (thus probably increasing the flux of incoming neutral reactants); on the other hand, some other experiments showed an increase in roughness with increasing pressure during plasma etching of Si in Cl_2 ⁸⁸ and SF_6 .⁹⁰ The smoothing with increased T_s has not so far been observed in experiments, which showed no T_s dependence of the roughness during plasma etching of Si in SF_6 at $T_s = -100$ to 40°C ⁸⁹ and in Cl_2 at $T_s = 20\text{--}200^\circ\text{C}$.⁷⁴

The decrease in roughness with increasing Γ_n^0/Γ_i^0 and T_s (not scaling with the etch rate) is attributed to increased chemical effects of neutral reactants (chemical and ion-enhanced etching reactions) on top of roughness features:⁷ the incident flux of neutral reactants is assumed to be smaller at the bottom than at the top of roughness features under shadowing effects of the feature, so the top of the features tends to etch faster than their bottom under conditions that the feature bottom starves for neutrals, which suppresses the evolution of surface roughness and ripples or causes surface smoothing. In effect, the smoothing with increased Γ_n^0/Γ_i^0 and T_s is less significant for 2D ripples at $\theta_i = 45^\circ$ and 75° than for 3D random roughness features at $\theta_i = 0^\circ$, because the shadowing effects for neutrals are lower for 2D than for 3D features.^{93–95} The smoothing with increased Γ_n^0/Γ_i^0 and T_s is also less significant at increased E_i , where the increased directionality of incoming ion fluxes is assumed to increase the flux of ions directly incident on the bottom of roughness features that starves for neutrals, and thus to effectively lower the effects of neutral shadowing.

These similarities would be no wonder between the effects of reduced ion reflection and those of increased neutral reactant flux and surface temperature, because the formation and evolution of surface roughness (and ripples) during plasma etching as well as IBS is assumed to be due to a competition between roughening (and rippling) and smoothing:⁷ the former is caused primarily by ion-enhanced etching reactions through ion bombardment, whose rates tend to be higher at the bottom of roughness features than at their top in the presence of ion reflection followed by re-impingement, unless the feature bottom starves for neutrals under shadowing; and the latter is by chemical effects of neutral reactants, which are more significant on top of roughness features under shadowing. In other words, the surface roughening (and rippling) occurs when the roughening dominates over smoothing through enhanced roughening and/or reduced smoothing, and the surface smoothing/non-roughening occurs when the smoothing dominates over roughening through reduced roughening and/or enhanced smoothing. Thus, in terms of smoothing, the reduced ion reflection (leading to reduced roughening) should be effectively equivalent to the increased neutral reactant flux and surface temperature (leading to enhanced smoothing).

This understanding gives an insight into the reason why the plasma-induced formation and evolution of roughness and ripples is so highly sensitive to the degree of ion reflection as in Figs. 1–3: small deviations of the reflection coefficient r_i

from unity result in smoothing/non-roughening (with no ripples at off-normal θ_i), which is due to the chemical or smoothing effects of neutral reactants that are incident preferentially on top of roughness features under the effects of shadowing. Then, the deviation of the critical r_i^* from unity should be much smaller (or the critical r_i^* is larger, closer to unity) with increasing the effects of smoothing and/or decreasing those of roughening; on the contrary, the deviation should be larger (or the critical r_i^* is much smaller than unity) with decreasing smoothing effects and/or increasing those of roughening (as in Fig. 3 for increased E_i). In effect, additional simulations made by varying r_i with different Γ_n^0/Γ_i^0 and T_s (for a given E_i and θ_i with $\Gamma_o^0/\Gamma_i^0 = 0$), not presented here, showed that the decrease in roughness (or the smoothing/non-roughening) with reduced ion reflection tends to be less significant at decreased Γ_n^0/Γ_i^0 and T_s : the lower the Γ_n^0/Γ_i^0 and the T_s , the smaller is the critical r_i^* giving the rms roughness decreased down to the low level of stochastic roughening. In addition, simulations made by varying r_i with different inhibitor oxygen-to-ion flux ratios ($\Gamma_o^0/\Gamma_i^0 = 0-0.5$,⁷³ for a given E_i and θ_i with $\Gamma_n^0/\Gamma_i^0 = 100$ and $T_s = 320$ K) further showed that the decrease in roughness with reduced ion reflection tends to be less significant at increased Γ_o^0/Γ_i^0 : the higher the Γ_o^0/Γ_i^0 , the smaller is the critical r_i^* , as expected from the effects of a small amount of incoming inhibitor flux that enhances the evolution of surface roughness or causes surface roughening (leading to the formation of columnar nanostructures like pillars at $\theta_i = 0^\circ$) through micromasking via local passivation layer formation on top of roughness features of Si under shadowing.^{5,7,18,27,73}

C. Effects of reduced ion reflection on ripple topography and dynamics

The effects of reduced ion reflection on ripple topography and dynamics would also be intriguing, since the surface features at intermediate off-normal $\theta_i = 45^\circ$ in Fig. 1(b) appear to exhibit the ripple wavelength increased as the ion reflection coefficient r_i is decreased (or equivalently, the smoothing is enhanced). Similar tendencies were also observed in ASCeM-3D simulations with $r_i = 1$, not presented here, through increasing the neutral reactant flux⁷³ and surface temperature.⁹² Moreover, in IBS studies, the enhanced smoothing has been known to lead to an increase in the wavelength of parallel-mode ripples (wave vector parallel, or wave crests/troughs perpendicular, to the direction of ion incidence) at lower beam fluxes and higher substrate temperatures:^{36,96,97} 0.8-keV Ar^+ beams onto Cu at $\theta_i = 70^\circ$ with $\Gamma_i^0 = (2.1-6.7) \times 10^{14} \text{ cm}^{-2} \text{ s}^{-1}$ and $T_s = 393-472$ K showed the ripple wavelength of $\lambda_r \approx 300-2000$ nm increasing with decreasing Γ_i^0 and increasing T_s ;⁹⁶ these experimental observations were reproduced by continuum models and MC simulations, suggesting that a competition between ion-induced roughening via sputtering and thermal smoothing via surface diffusion of mobile defects is responsible for the enhanced smoothing.^{36,96,97}

Figure 4 shows the time evolution of surface features of Si (top view, $50 \times 50 \text{ nm}^2$) during etching in Cl_2 plasma

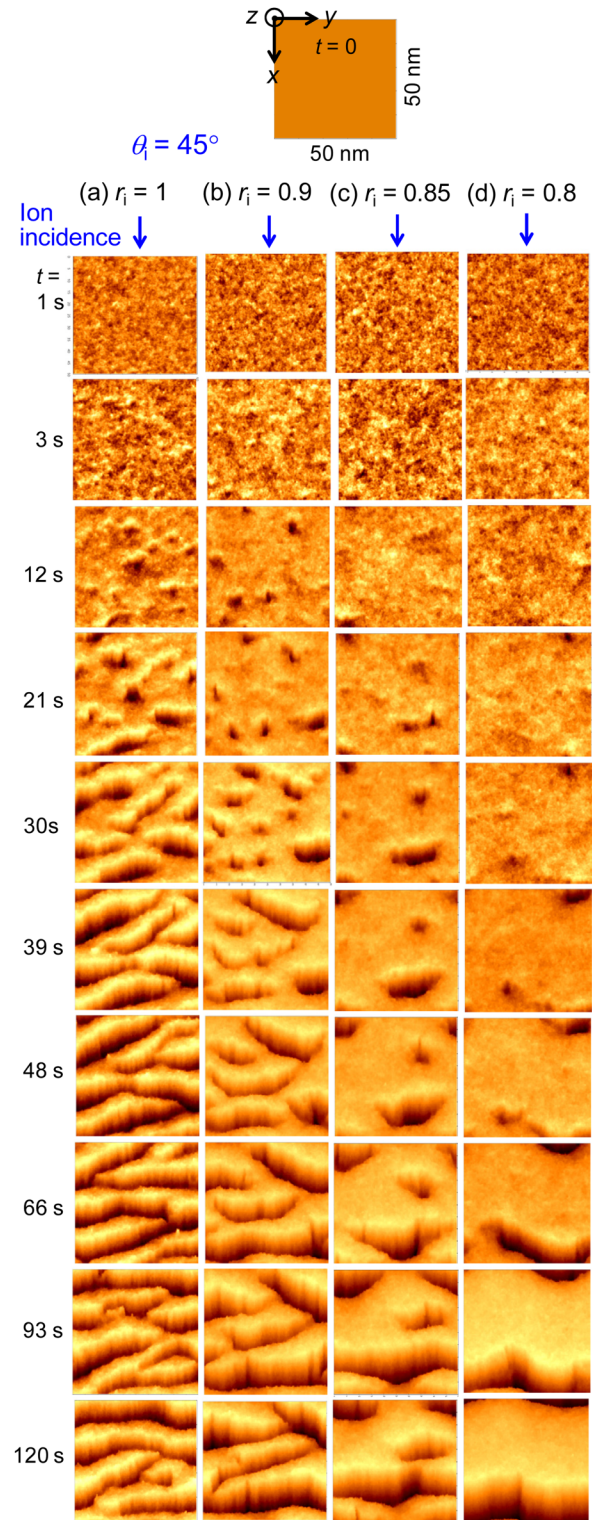


FIG. 4. Time evolution of surface features of Si (top view, $50 \times 50 \text{ nm}^2$) during etching in Cl_2 plasma ($0 \leq t \leq 120$ s) for different ion reflection coefficients of $r_i =$ (a) 1, (b) 0.9, (c) 0.85, and (d) 0.8 with $E_i = 100$ eV at $\theta_i = 45^\circ$, extracted from simulations as in Fig. 1(b). The coordinate system (x , y , z) used in the model is also inset here (as in Fig. 1), and the time evolution of the etch rate and rms surface roughness concerned here is the same as in Figs. 2(c) and 2(d). Note that the surface features shown are their snapshots taken every $T = W/v_r \approx 9$ s and/or its multiples (18 and 27 s) in the range $3 \leq t \leq 120$ s, where $W = 50$ nm is the side length of the square substrates being simulated, and $v_r \approx 5.5$ nm/s is the traveling velocity of surface features in the direction of ion incidence (or in the x -direction) during etching.

($0 \leq t \leq 120$ s) for different $r_i = 1, 0.9, 0.85$, and 0.8 with $E_i = 100$ eV at $\theta_i = 45^\circ$, extracted from simulations as in Fig. 1(b). The corresponding representative side or cross-sectional views of surface features are shown in Fig. 5, where the line of sight is perpendicular to the direction of ion incidence (or in the y -direction), and the simulation domain shown is 2 nm in width (in the y -direction at around the x axis at $y = 22, 12, 40$, and 40 nm for $r_i = 1, 0.9, 0.85$, and 0.8 , respectively). The coordinate system (x, y, z) used in ASCeM-3D is inset in Fig. 4 (also in Fig. 1), and the time evolution of the etch rate and rms surface roughness concerned is the same as earlier in Figs. 2(c) and 2(d). Note that in Figs. 4 and 5, the top- and side-view images of surface features shown are their snapshots taken every 9 s and/or its multiples (18 and 27 s) in the range $3 \leq t \leq 120$ s. In the present situations, the surface features or ripple patterns move or travel laterally with the velocity of approximately $v_r \approx 5.5$ nm/s on etched surfaces in the direction of ion incidence (or in the x -direction) continuously after the start of etching, as seen later in Figs. 6 and 7; thus, these snapshot

images can be considered to be those which observers at rest on the traveling surface features would see every $T = W/v_r \approx 9$ s and/or its multiples, where $W = 50$ nm is the length of one side of the square substrates being simulated.

Simulations exhibit that soon after the start of etching, the initially flat surfaces (RMS = 0) are roughened randomly with small concavo-convex roughness features at any θ_i (stage I), while the evolution of surface features thereafter depends largely on θ_i .^{7,72,73} at intermediate $\theta_i = 45^\circ$, small roughness features tend to merge with each other with time to form concavo-convex features enlarged laterally as well as vertically (stage II), which evolve further into pronounced ripple structures with wave crests/troughs elongated perpendicular to the direction of ion incidence (stage III). The stages I, II, and III correspond approximately to the respective time ranges in Fig. 2(d) exhibiting a little increase in RMS (< 0.3 , the low level of stochastic roughening) soon after the start of etching ($0 < t < t_1 \approx$ a few seconds irrespective of r_i), the subsequent slight increase in RMS (< 0.5 , $t_1 < t < t_2$), and then a significant increase in RMS (> 2) thereafter

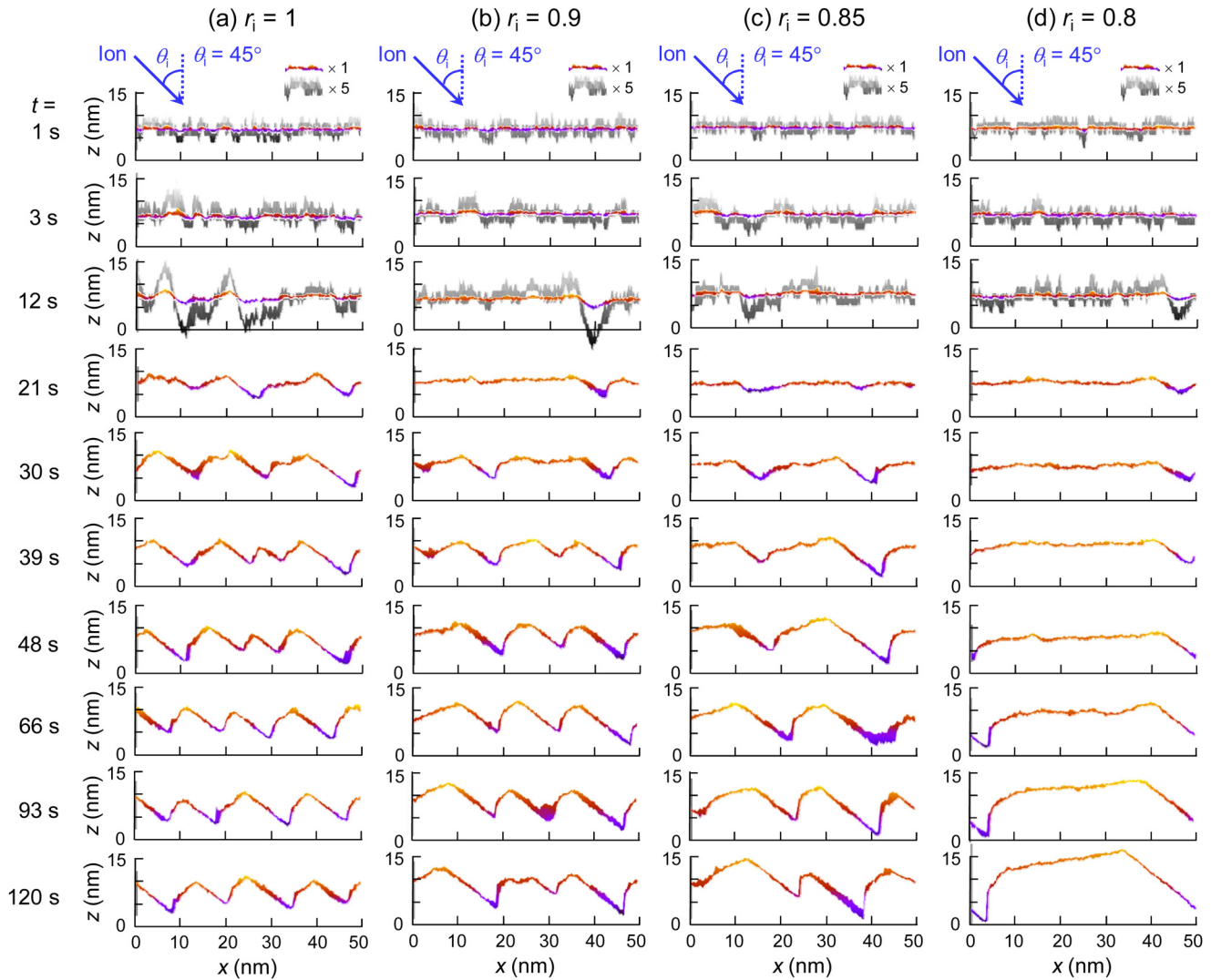


FIG. 5. Time evolution of representative side or cross-sectional views of surface features of Si during etching ($0 \leq t \leq 120$ s) for different $r_i =$ (a) 1, (b) 0.9, (c) 0.85, and (d) 0.8 with $E_i = 100$ eV at $\theta_i = 45^\circ$, corresponding to the preceding Fig. 4. The line of sight here is perpendicular to the direction of ion incidence (or in the y -direction), and the simulation domain shown is 2 nm in width (in the y -direction at around the x axis at $y = 22, 12, 40$, and 40 nm for $r_i = 1, 0.9, 0.85$, and 0.8 , respectively).

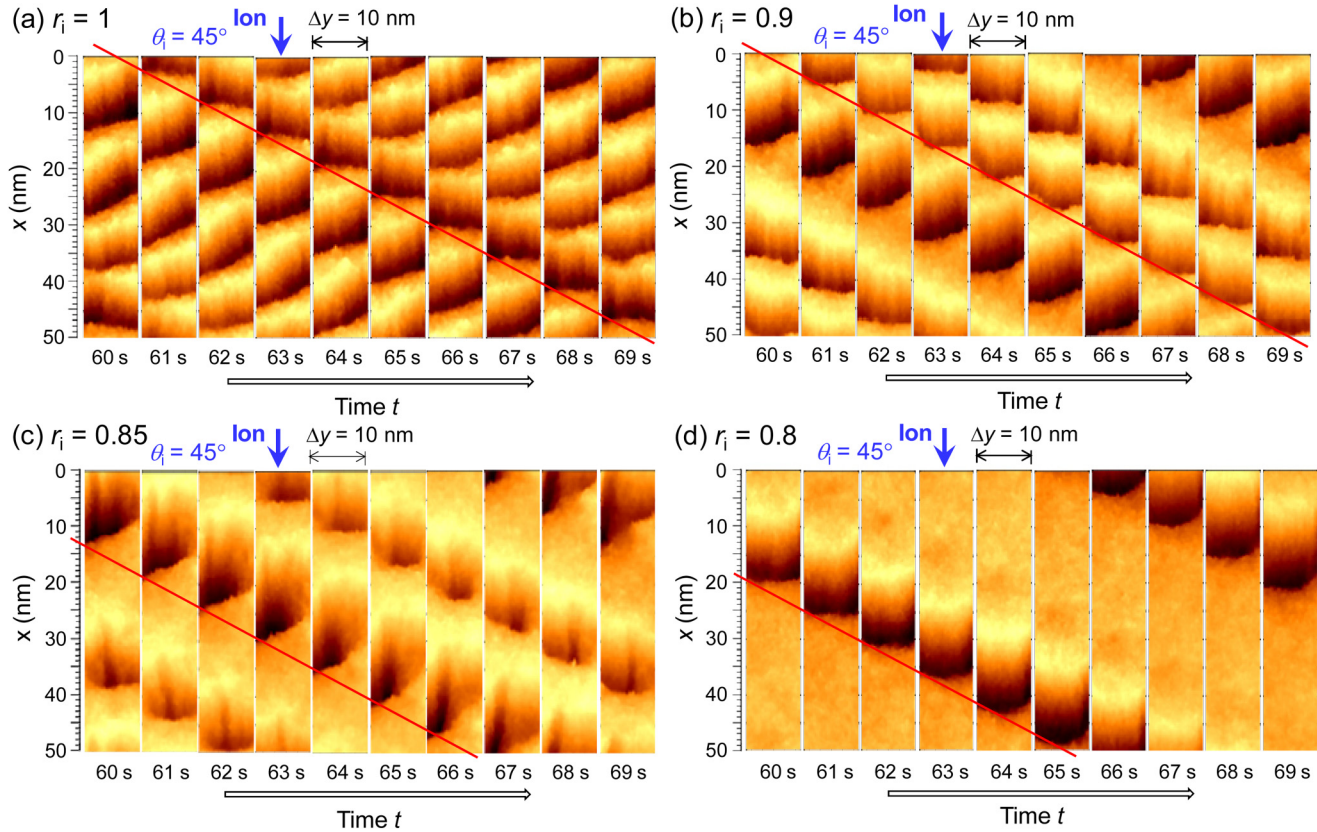


FIG. 6. Time sequence of surface features or ripple patterns of Si at $t=60\text{--}69$ s during etching in Cl_2 plasma for different ion reflection coefficients of $r_i =$ (a) 1, (b) 0.9, (c) 0.85, and (d) 0.8 with $E_i = 100$ eV at $\theta_i = 45^\circ$ (top view, $\Delta x \times \Delta y = 50 \times 10$ nm² of the total area 50×50 nm² at around the x axis at $y = 22, 12, 40$, and 40 nm, respectively), extracted from simulations as in Fig. 1(b). The red straight lines in the respective figures are for guiding the eyes only, showing that given surface feature points move or travel laterally at a velocity $v_r \approx 5.5$ nm/s across the surfaces being etched in the direction of ion incidence (or in the x -direction).

($t_2 < t < t_3$). The critical times concerned are longer for decreased r_i , being $t_2 \sim 10, 20, 20$, and 30 s and $t_3 \sim 35, 60, 90$, and 90 s for $r_i = 1, 0.9, 0.85$, and 0.8 , respectively.

The evolution reaches quasi-steady state at around the end of stage III, exhibiting parallel-mode ripples characterized by approximately four, three, two, and one crest(s)/trough(s) for $r_i = 1, 0.9, 0.85$, and 0.8 , respectively; in other words, the distance between crests/troughs (crest-to-crest/trough-to-trough) is larger, or the ripple wavelength is longer, for decreased r_i . For further decreased $r_i < 0.8$ (or $r_i \leq r_i^* \approx 0.75$), the surfaces remain smooth at the low level of stochastic roughening during etching after only a little increase in roughness at the initial stage I, without the subsequent evolution of roughness and ripples. Note that the ASCeM-3D obeys periodic boundary conditions in horizontal x - and y -directions (as mentioned in Sec. II), giving the surface heights/depths $z(x=0, y) = z(x=W, y)$ and $z(x, y=0) = z(x, y=W)$ at boundaries; thus, in simulations, the number of wave crests/troughs formed should be an integer ($0, 1, 2, 3, 4, \dots$), or the ripple wavelength should be nominally a fraction ($\infty, 1, 1/2, 1/3, 1/4, \dots$) of the simulation domain size $W = 50$ nm. The smoothing/non-roughening for decreased $r_i < 0.8$ at $\theta_i = 45^\circ$ may be attributed partly to the wavelength of the ripples to be formed that would exceed the size limit W .

The top- and side-view images of Figs. 4 and 5 further exhibit that at stage II ($t_1 < t < t_2$), the roughness features

enlarged at intervals are fewer in number density for decreased r_i ; this in turn leads to the number of wave crests/troughs elongated that is smaller (or to the ripple wavelength that is longer) for decreased r_i at stage III ($t_2 < t < t_3$), when the roughness features enlarged are merged laterally (preferentially in the direction perpendicular to that of ion incidence) to form pronounced parallel-mode ripple structures. At stage III in side-view images of Fig. 5, the number of crests/troughs looks as if it increased (or the wavelength decreased) with time, owing to the lateral merging of roughness features, except the case $r_i = 0.8$ where it remains one. Thus, we consider that the reduced ion reflection (or equivalently the enhanced smoothing) suppresses the formation of roughness features enlarged at intervals, which in turn leads to the decrease in number of wave crests/troughs elongated, or to the increase in wavelength of the ripples formed, for decreased r_i .

Figure 6 shows the time sequence of surface features or ripple patterns of Si at $t=60\text{--}69$ s during etching in Cl_2 plasma for different $r_i = 1, 0.9, 0.85$, and 0.8 with $E_i = 100$ eV at $\theta_i = 45^\circ$ (top view, $\Delta x \times \Delta y = 50 \times 10$ nm² of the total area 50×50 nm² at around the x axis at $y = 22, 12, 40$, and 40 nm, respectively), extracted also from simulations as in Fig. 1(b). The corresponding representative side or cross-sectional views of surface features or ripple structures are shown in Fig. 7, where the figure properties (the line of sight and the simulation domain shown) are the same as in Fig. 5. A series

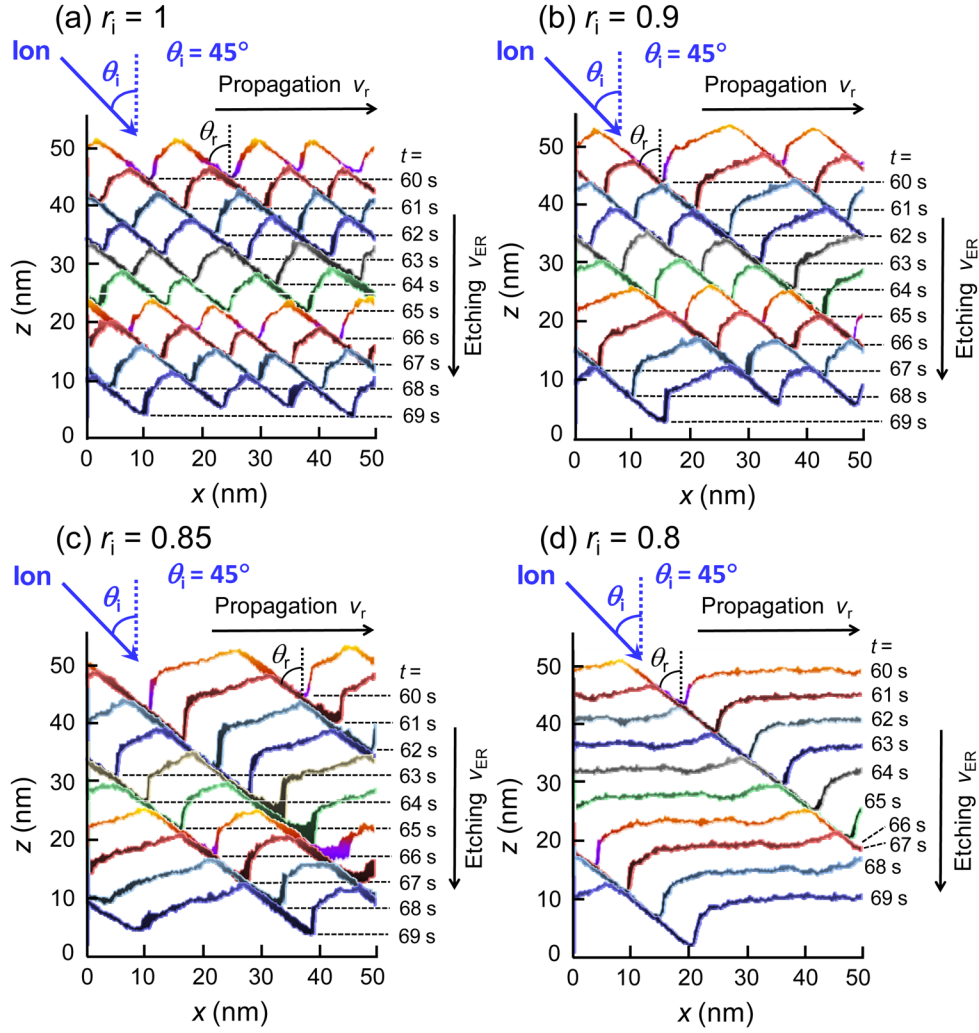


FIG. 7. Time sequence of representative side or cross-sectional views of surface features or ripple structures of Si at $t = 60$ – 69 s during etching for different r_i = (a) 1, (b) 0.9, (c) 0.85, and (d) 0.8 with $E_i = 100$ eV at $\theta_i = 45^\circ$, corresponding to the preceding Fig. 6. The figure properties (the line of sight and the simulation domain shown) are the same as in Fig. 5. The surface features or ripples move or travel laterally at $v_r \approx 5.5$ nm/s across the surfaces being etched in the direction of ion incidence (or in the x -direction), with the surfaces being moved vertically downward (or in the negative z -direction) at $v_{ER} = ER/60 \approx 4.3$ nm/s through etching; then, the ripple angle, defined as the angle between the downward slope of the ripple and the surface normal of substrates, is correlated with the ion incidence angle as approximately $\theta_r \approx \theta_i = 45^\circ$.

of these time-varying top- and side-view images indicate that the surface features or ripples [characterized by four, three, two, and one wave crest(s)/trough(s) for $r_i = 1, 0.9, 0.85$, and 0.8 , respectively] move or travel laterally across the surface being etched in the direction of ion incidence (or in the x -direction), with the surfaces being moved vertically downward (or in the negative z -direction) through etching. The propagation of parallel-mode ripples in the ion incidence direction has so far been observed in several IBS experiments^{70,98–101} (although there is a recent exception),¹⁰² which is opposite to the predictions of continuum models⁷¹ and MC simulations^{97,103–106} of the ion beam-induced ripple formation. This discrepancy may suggest that the ion-induced ripple formation and evolution and its topography and dynamics are largely affected by the ion reflection followed by re-impingement on feature surfaces, the effects of which have not been included in IBS models and simulations,^{36,71,97} although they include surface processes such as ion-induced sputtering and surface diffusion of defects.

The average traveling velocity, derived from the time rate of change of the position of given surface feature points (e.g., those of wave crests/troughs), is approximately $v_r \approx 5.5$ nm/s almost independent of r_i , which is a little higher than the vertically downward $v_{ER} = ER/60 \approx 4.3$ nm/s through etching; as a result, similar surface features or ripple patterns are seen to occur in cycles with a period of $T = W/v_r \approx 9$ s at quasi-steady state in Figs. 4–7 [note that the images in Fig. 1(b) are those seen at $t = 60$ s between $t = 48$ and 66 s in Fig. 4]. In simulations, the traveling velocity v_r as well as the descending velocity v_{ER} was observed to remain almost unchanged during etching ($0 < t < 120$ s). Similar propagation velocities of parallel-mode ripples have been reported in IBS experiments as mentioned above using 30-keV Ga^+ beams:^{70,98–101} typically (with $\Phi \approx 1 \times 10^{18}$ ions/cm²), $v_r \approx 0.33$ nm/s ≈ 44 nm/(10^{17} ions cm⁻²) on Si at $\theta_i = 30^\circ$ ($\lambda_r \approx 100$ nm);⁹⁸ $v_r \approx 0.75 v_0$ on SiO_2 at $\theta_i = 45^\circ$ ($\lambda_r \approx 200$ nm, $z_r \approx 5$ nm, v_0 the erosion velocity);⁹⁹ $v_r \approx 2.3$ nm/s ≈ 66 nm/(10^{17} ions cm⁻²) on $\text{Cd}_2\text{Nb}_2\text{O}_7$ pyrochlore at

$\theta_i = 45^\circ$ ($\lambda_r \approx 400$ nm, $z_r \approx 150$ nm);¹⁰⁰ and $v_r \approx 77$ nm/(10^{17} ions cm^{-2}) on SiO_2 at $\theta_i = 52^\circ$ ($\lambda_r \approx 450$ nm, $z_r \approx 130$ nm).^{70,101} Note that the present $v_r \approx 5.5$ nm/s corresponds to $\approx 1.3v_{\text{ER}}$ and to ≈ 55 nm/(10^{17} ions cm^{-2}).

The amplitude (crest-to-trough) as well as wavelength (crest-to-crest/trough-to-trough) of the ripples is derived from side-view images as in Figs. 5 and 7, indicating that at quasi-steady state, the amplitude is approximately in the range $z_r \approx 6$ –12 nm (increasing slightly with decreasing r_i from 1 to 0.8). The increase of the amplitude z_r with r_i is not so significant as compared to that of the wavelength which is nominally in the range $\lambda_r \approx (1/4 - 1)W = 12.5$ –50 nm; in a sense, the tendency of z_r reflects that of the rms roughness depending on r_i [Figs. 2(b) and 3(b)]. It is further noted that at quasi-steady state, the ripple angle, defined as the angle between the downward slope of the ripple and the surface normal of substrates, is correlated with the ion incidence angle as approximately $\theta_r \approx \theta_i = 45^\circ$ irrespective of r_i ; in other words, the downward-sloping surfaces of the ripple are nearly parallel to the ion incidence, in such a way that no shadowing effects of the feature occur for incoming ions.^{37,48} On the other hand, the upward-sloping surfaces consist of lower parts having a steep slope followed by upper ones having a relatively gentle one, where the latter are more gently sloped for decreased r_i with the ripple wavelength being increased. The steep slope of lower parts of the upward surfaces is attributed to faceting caused by re-impingement of ions reflected from the adjacent downward surfaces on incidence in addition to direct incidence of incoming ions,⁵³ while the gentle slope of upper parts is to faceting caused only by incoming ions directly incident thereon.

For $r_i = 1$, the slopes of upper and lower parts of the upward-sloping surfaces tend to be similar, approximately perpendicular to the ion incidence, exhibiting the formation of well-defined sawtooth-like ripples ($\lambda_r \approx 12.5$ nm, $z_r \approx 6$ nm for 100-eV Cl^+ on Si at $\theta_i = 45^\circ$) as observed in our plasma etching⁷⁸ as well as IBS^{66–69,100,107–109} experiments: typically, λ_r (nm)/ z_r (nm) $\approx 62/18$ for 0.5-keV Cl_2^+ (in practice, probably SiCl^+ as discussed earlier in Sec. III A) on Si at $\theta_i \approx 40^\circ$,⁷⁸ $\approx 350/55$ for 0.5-keV Ar^+ on Si at 70° ,^{66,67} $\approx 60/5$ for 1.2-keV Xe^+ on Si at 75° ,⁶⁸ $\approx 40/5$ for 2-keV Kr^+ on Si at 75° ,⁶⁹ $\approx 400/150$ for 30-keV Ga^+ on $\text{Cd}_2\text{Nb}_2\text{O}_7$ at 45° ,¹⁰⁰ $\approx 500/40$ for 20-keV Ga^+ on diamond at 80° ,^{107,108} and $\approx 250/50$ for 12-keV N^+ on Si at 60° .¹⁰⁹ Such characteristic sawtooth-like (or terrace-like) profiles with $\theta_r \approx \theta_i$ are assumed to reflect the limiting condition for the formation of parallel-mode ripple structures,^{37,48,78} which causes shadowing not to occur and faceting to be fully developed.

The wavelengths and amplitudes of fully developed sawtooth-like ripples with $\theta_r \approx \theta_i$ for intermediate off-normal $\theta_i = 40^\circ$ – 60° , observed in the present simulations, plasma etching,⁷⁸ and IBS experiments^{100,109} at different $E_i = 0.1$ –30 keV, appear to scale with the ion energy as $\lambda_r \sim E_i^p$ and $z_r \sim E_i^q$ with $p, q \approx 0.6$.⁷⁸ The ripple wavelength and amplitude in IBS are appreciated to rely on ion energy, incidence angle, and fluence,^{36–39} and so this scaling may also suggest that the ion-induced formation and evolution of ripple structures is largely affected not only by the ion shadowing and

faceting^{40,48} but also by the ion reflection followed by re-impingement on feature surfaces.⁵³ In simulations, sawtooth-like profiles tend to be collapsed with decreasing r_i , where the reduced ion reflection suppresses the evolution of roughness features or ripple structures to both vertical and lateral extent; in effect, if sawtooth-like profiles were fully developed for decreased r_i , the roughness height/depth or the ripple amplitude would be more significantly increased, corresponding to the increase in wavelength.

Figure 8 shows the wavelength λ_r and amplitude z_r of parallel-mode ripples of Si as a function of ion reflection coefficient in the range $r_i = 1$ –0.5, together with the rms surface roughness [Figs. 2(b) and 2(d)], obtained through simulations of Si etching in Cl_2 plasma with $E_i = 100$ eV at $\theta_i = 45^\circ$ as in Fig. 1(b). Also shown are the corresponding traveling velocity v_r of the ripples, together with the etch rate or the surface descending velocity $v_{\text{ER}} = \text{ER}/60$ [Figs. 2(a) and 2(c)]. These are the data at around $t = 60$ s after the start of etching, providing a summary of the ripple topography and dynamics at quasi-steady state depending on r_i , as seen in top- and side-view images of surface features and/or ripple structures of Figs. 4–7. Note that in simulations, the surface features were observed to move or travel laterally on etched surfaces in the direction of ion incidence during etching irrespective of $r_i = 1$ –0, although the roughness is substantially reduced and ripples fade away for decreased $r_i < 0.8$. As the coefficient r_i is decreased, the ripple wavelength λ_r increases significantly, while the increase in amplitude z_r with r_i is not so significant as compared to that of the wavelength; as a result, sawtooth-like ripple profiles, pronounced for $r_i = 1$, tend to be collapsed with decreasing r_i , although the ripples move or travel laterally across the surfaces being etched in the direction of ion incidence with the velocity v_r almost independent of r_i . This is consistent with IBS studies showing that the enhanced smoothing leads to an increase in the wavelength of parallel-mode ripples at lower beam fluxes

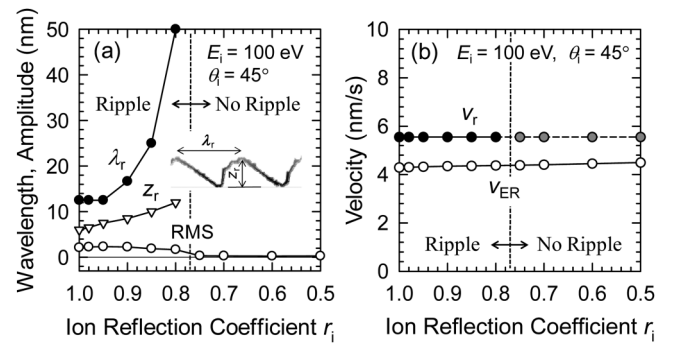


FIG. 8. (a) Wavelength λ_r and amplitude z_r of parallel-mode ripples of Si as a function of ion reflection coefficient in the range $r_i = 1$ –0.5, together with the rms surface roughness [Figs. 2(b) and 2(d)], obtained through simulations of Si etching in Cl_2 plasma with $E_i = 100$ eV at $\theta_i = 45^\circ$ as in Fig. 1(b). Also shown are the corresponding (b) traveling velocity v_r of the ripples, together with the etch rate or the surface descending velocity $v_{\text{ER}} = \text{ER}/60$ [Figs. 2(a) and 2(c)]. The data shown are those at around $t = 60$ s after the start of etching, which are the averages extracted from several runs of the simulation except for nominal values of λ_r . These provide a summary of the ripple topography and dynamics at quasi-steady state depending on r_i , as seen in top- and side-view images of surface features and/or ripple structures of Figs. 4–7.

and higher substrate temperatures.^{36,96,97} These results indicate that the reduced ion reflection affects significantly the ripple topography and dynamics induced by ion bombardment, which is equivalent to effects of the enhanced smoothing, although the smoothing in plasma etching is due to chemical effects of neutral reactants that are involved under shadowing.

IV. CONCLUSIONS

We have investigated the origin of plasma-induced surface roughening and ripple formation, based on MC-based ASCeM-3D simulations of plasma-surface interactions and feature profile evolution during Si etching in Cl-based plasmas. The model took into account the Cl^+ ion reflection (and/or its penetration into substrates) through calculating the momentum and energy conservation in successive two-body elastic collisions with substrate Si atoms every ion incidence (not assuming specular reflection), thus giving internally the reflection probability $R_i(E_i, \theta)$ along with the distributions of reflected energies and angles. In this study, the “reflection coefficient r_i ” was further introduced in the model ($0 \leq r_i \leq 1$), to quantitatively analyze the role and effects of ion reflection from microstructural feature surfaces on incidence. The coefficient r_i represents the fraction of ions incident on surfaces with the reflection/penetration calculation scheme turned on, and so the r_i is a measure of the reflection probability \mathcal{R}_i for impacts of an ion species onto Si surfaces (a second condition of interest) relative to that R_i for Cl^+ impacts (base condition): $\mathcal{R}_i(E_i, \theta) = r_i \times R_i(E_i, \theta)$, where $0 \leq r_i$, R_i , $\mathcal{R}_i \leq 1$, and θ is the local incidence angle on feature surfaces.

Simulations with ion incident energies in the range $E_i = 20\text{--}500$ eV for incidence angles of $\theta_i = 0^\circ$, 45° , and 75° onto substrate surfaces showed that as the coefficient r_i is slightly decreased from unity, the roughness decreases substantially and the ripple formation fades away: the roughness remains at the low level of stochastic roughening ($\text{RMS} < 0.3\text{--}0.5$ nm, no pronounced ripples at off-normal θ_i) during etching for decreased $r_i \leq r_i^*$, where the critical $r_i^* \approx 0.95$, 0.75 , and 0.8 at $\theta_i = 0^\circ$, 45° , and 75° , respectively, e.g., for $E_i = 100$ eV. This clearly indicates the crucial role of ion reflection from feature surfaces on incidence at microscale during plasma etching: it is indispensable in plasma-induced surface roughening and rippling, and their degree relies significantly on the reflectivity of ions. The decrease in roughness (or smoothing/non-roughening) with reduced ion reflection tends to be less significant at increased E_i and θ_i : the higher the E_i and the θ_i , the smaller is the critical r_i^* ($\approx 0.95\text{--}0.75$) or the larger is the deviation of the critical r_i^* from unity.

The critical value $r_i^* \approx 0.95\text{--}0.8$ at normal $\theta_i = 0^\circ$ is consistent with our experiments of ICP plasma etching of Si in Cl_2 exhibiting the roughening-smoothing transition at increased $E_i > 200\text{--}300$ eV,^{7,74,75} where the predominant ion flux was found to change from reactive $\text{Cl}_2^+/\text{SiCl}_3^+$ (base condition) to depositive SiCl^+ (a second condition); in effect, the reflection probabilities tend to be lower for the latter than for the former, according to our MD simulations.^{7,77} Moreover, the smaller $r_i^* \approx 0.8\text{--}0.75$ at off-normal $\theta_i = 45^\circ$ and 75° (as compared to $r_i^* \approx 0.95\text{--}0.8$ at $\theta_i = 0^\circ$) is also

consistent with our experiments of Si etching in Cl_2 plasmas using sheath control plates,⁷⁸ demonstrating the ripple formation under conditions of $E_i \approx 500$ eV with $\theta_i \approx 40^\circ$ and 80° (though in the smoothing mode for $\theta_i = 0^\circ$, where the predominant ions were SiCl^+ with relatively low reflectivity). Similarity of the smoothing/non-roughening behavior between the reduced ion reflection and the increased reactant flux⁷³ and surface temperature⁹² in simulations gives an insight into the reason why the plasma-induced formation and evolution of roughness and ripples is highly sensitive to the degree of ion reflection: small deviations of the critical r_i^* from unity are ascribed to chemical or smoothing effects of neutral reactants incident preferentially on top of roughness features under the effects of shadowing, which compete with the roughening and rippling caused by ion bombardment.

There would be some other experimental observations of plasma-induced surface roughening and rippling interpreted similarly according to the results of this study, where the understanding of plasma conditions is required, including the predominant ion species and their reflection probabilities on surfaces concerned. In any case, the cause-and-effect relationships between the ion reflection and the surface roughness in plasma etching have not been fully understood due to their complexity; in effect, the former crucially affects the latter as in this study, while in turn, the former (reflection probability, distributions of reflected energies and angles) is affected by the latter as in earlier MD studies of Helmer and Graves.¹¹⁰

Simulations further showed that at intermediate off-normal $\theta_i = 45^\circ$, the ripple wavelength increases significantly from $\lambda_r \approx 12.5$ to 50 nm as the coefficient is decreased from $r_i = 1$ to 0.8 for a given $E_i = 100$ eV, while the increase in amplitude with r_i is not so significant as compared to that of the wavelength; as a result, sawtooth-like ripple profiles pronounced for $r_i = 1$ tend to be collapsed with decreasing r_i , although the ripples move or travel laterally across the surfaces being etched in the direction of ion incidence with the velocity $v_r \approx 5.5$ nm/s almost independent of r_i . This is consistent with IBS studies showing that the enhanced smoothing leads to an increase in the wavelength of parallel-mode ripples at lower beam fluxes and higher substrate temperatures;^{36,96,97} moreover, the ripple propagation is also consistent with experimental observations during IBS,^{70,98–101} and sawtooth-like (or terrace-like) ripple profiles are with observations in plasma etching⁷⁸ and IBS^{66–69,100,107–109} experiments. Thus, we consider that the ripple topography and dynamics induced by ion bombardment are also significantly affected by the ion reflection followed by re-impingement on feature surfaces, where the reduced ion reflection is effectively equivalent to the enhanced smoothing, although the smoothing in plasma etching is caused by chemical effects of neutral reactants that are involved under shadowing.

The results and understandings obtained in this study would be intriguing and informative for ion beam- as well as plasma-surface interactions, providing insight into the mechanisms underlying a number of experimental observations of ion-induced surface morphology evolution. Modeling and simulation studies like the present one, taking into account self-consistently the effects of ion reflection, should be useful

and indispensable to unravel the complex cause-and-effect relationships between it and surface roughness, and to quantitatively understand what plasma and/or beam conditions lead to what degree of surface roughening and rippling.

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